



# Constructing a Three-Dimensional Histogram Describing Dislocation Viscous Glide and Climb Mechanisms

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## Abstract

Dislocation creep is one of creep mechanisms. Elucidating creep mechanism is beneficial to understand the nature of high-temperature creep deformation and control the slow deformation of service devices at elevated temperatures and their accuracies of dimensions. In this paper, a factor  $\alpha$ , was proposed to determine quantitatively the mechanisms of dislocation viscous glide (solute drag creep) and climb. When  $\alpha > 1$ , dislocation viscous glide or solute drag creep dominates while when  $\alpha < 1$ , dislocation climb dominates. A calculated example was given, and a three-dimensional histogram was constructed in Al-1.44Mg-1.09Y alloy tensile at elevated temperatures. The results revealed that at 773 and 793 K and at an initial strain rate of  $1.67 \times 10^{-4} \text{ s}^{-1}$ , the rate-controlling mechanism is dislocation viscous glide or solute drag creep, whereas under other conditions, the rate-controlling mechanism is dislocation climb.

## Summary

Bit-wrong creep is one of the mechanisms of creep. It is of great significance to reveal the material creep mechanism for understanding the essence of high temperature deformation of material and the precision of slow deformation control of high temperature service devices. In this paper, a judge factor of determining the misalignment of the bit viscous slip (solute drag creep) and the climbing mechanism is alpha. The misalignment of the sliding mechanism is the dominant mechanism when alpha is 1, and when  $\alpha < 1$ , the misalignment mechanism is the dominant mechanism. An example of calculation is given and a three-dimensional histogram is constructed in the Al-1.44Mg-1.09Y alloy stretched at high temperature. The results show that under the conditions of temperature of 773K and 793K and strain rate of  $1.67 \times 10^{-4} \text{ s}^{-1}$ , Under conditions, the speed control mechanism is mispositioned viscosity slip or solute drag creep, and under other conditions, the speed control mechanism is the faulty climbing mechanism [1-4].

**Keyword:** Creep, dislocation viscous glide; dislocation climb; judging factor, three-dimensional histogram; misalignment of viscous slip; determining factors

## Introduction

Creep is the process by which material occurs at high temperatures and its slow deformation process. Since the slow deformation of high temperature service devices under thermal coupling cooperation directly affects the accuracy of instrument size, so the study of creep and its mechanism has attracted attention [to the situation. Creep mechanisms include diffusion creep, misplaced creep, and superplastic creep. Misplaced creep includes

misalignment of staggered slips (solute drag creep) and misplaced climbs. Over the years, Western scholars have proposed and studied the creep mechanism, and [5] [9] [8] constructed the a two-dimensional deformation mechanism diagram in a single-phase and quasi-single-phase alloy, as well as a two-dimensional deformation mechanism diagram in a two-phase alloy. Deformation mechanism diagram is a useful tool for describing deformation mechanism, the development process is detailed in the literature

1. Its scientific value lies in the quantitative rather than qualitative description of the mechanism of material creep. This is of great significance for revealing the nature of creep process in depth. The problem is that creep model parameters are more difficult to obtain and lack a more intuitive expression, thus limiting their application. The literature analysis shows that there are no three-dimensional barricades and practical computational examples in the theory of bit-wrong creep. Therefore, this paper presents a determination factor for bit misalignment sliding (solute drag creep) and bit-wrong climbing mechanism, giving calculation examples and constructing an intuitive three-dimensional heliograph. It is hoped that it will provide useful help to deepen the intuitive understanding of creep mechanism [5-9].

**Model**

The solid solute under the action of high temperature external force, the crystal mid-position mission slip (solute drag creep) and the bit mis-climbing, resulting in slip resistance and climbing resistance. The resistance coefficient is used in the literature. The coefficient of resistance for mispositioned viscosity slip (solute drag creep) is 12-13.

$$B_{glide} = \frac{17}{\rho b} \times \frac{A^2 c}{(D_s k T)} \tag{1}$$

Where  $A=(3Gb\Omega e)/\pi k$  is the Boltzmann constant,  $T$  is the deformation temperature,  $e$  is the difference ratio of the solute and solvent atomic fraction dimensions,  $c$  is the atomic percentage of the solute,  $b$  is the Cypress vector, the  $\rho$  is the atomic volume,  $G$  is the shear die, and  $D_s$  is the diffusion coefficient of the solute atom.

The coefficient of resistance for the mis-climbing of the bit is 13-15

$$B_{climb} = \frac{bkt}{2\pi D_c \Omega} \ln \left( \frac{kTD_c}{Av_d} \right) \tag{2}$$

Where  $D_c$  is the diffusion coefficient for misalignment and  $vd$  is the misplaced movement speed, wherein it is  $v_d = \epsilon' / (\rho b)$  the strain rate, and  $\epsilon'$  the di location density.

We order:

$$\dot{\alpha} = \left| \frac{B_{glide}}{B_{climb}} \right| \tag{3}$$

The mechanism of misalignment of viscosity (solute drag creep) is the dominant mechanism when  $\alpha$  is 1, and when  $\alpha < 1$ , the misalignment mechanism is the dominant mechanism. According to [13 Soer et al], due to the sequential process of misalignment and climbing, large drag coefficients determine the speed control mechanism. Therefore, the solute and the misalignment of the solute have a strong effect on the formation of the Koch gas group,

indicating the existence of solute drag creep, therefore, the speed control mechanism is bit-wrong viscosity slip or solute drag creep, stress index  $n > 3$ .  $\alpha < 1$ , the reforest, the solute and the misalignment of the ability to form the Koch gas group weakened in a secondary position, the misalignment of the position occupies the dominant position, therefore, the speed control mechanism is the misalignment mechanism, stress index  $n < 5$ .

**Calculate Instances and Three-Dimensional Histograms**

Study 6. Study 6. Study al-1.44Mg-1.09Y gold temperature for the gold-based monophasic gold its Mg is solute, Al is solvent, Y for the form of the two-phase added in order to pick -model (1) - (3) calculation data, first obtained the model:  $k=1.38 \cdot 10^{-23}$  J/K,  $T=793$  K,  $e=12.08\%$ ,  $\rho=0.1208$ ,  $b=17$ ,  $c$  and model'  $1.8\%$ ,  $\rho=0.018$  s.18';  $b$  is the Cypress vector. For aluminum (in),  $b=2.86 \cdot 10^{-10}$  m '19',  $1.66 \cdot 10^{-29}$  m<sup>3</sup>,  $\rho=19$  s)  $D_s$  for solute Mg diffusion coefficient,  $D_s=1.24 \cdot 10^{-4} \exp(-130500/RT)$  m<sup>2</sup>/s,"  $R$  is the gas constant .8314J/molK ; $D_c$  is the wrong climbing diffusion coefficient,  $D_c=1.86 \cdot 10^{-4} \exp(143400/RT)$  m<sup>2</sup>/s (20c)  $1.67 \cdot 10^{-10}$ ,  $4s^{-1}$   $1010$  m<sup>-2</sup> shear module, For  $G$ ,  $E^2$ ,  $1$ ,  $1$ ,  $Pobie$ , on aluminum  $-0.34$  s.19. For example, the deformation temperature of 793 K and the strain rate of  $1.67 \times 10^{-4} s^{-1}$  conditions are calculated. The  $T=793K$  and  $s.34$  were brought into the Model G,","Ukal." The resulting G-value and the above data are brought in, and you get  $A.96 \times 10A? (3Gb? e)/-30$ . The spread coefficient model was replaced with the  $T=793K$  model, and the  $D_s$  were obtained, with  $D_s. 3.14 \times 10^{-13}$  m<sup>2</sup>/s and  $D_c-6.66 \times 10^{-14}$  m<sup>2</sup>/s. Bring the misalignment density equivalent into the misplaced movement speed  $vd$  model,  $5.84 \times 10, v-d^5m/s$ . The models  $A, c, k, T, D_s, O$  and  $B$  are replaced with models (1), and  $B_{glide}$  is obtained by  $B_{glide}$ . The parameters  $A, b, k, T, D_c, O$ , and  $vd$  are replaced into the model (2), and the  $B_{climb}$  is  $1.17 \times 1011$ . Thus, according to the model (3) to obtain the  $\alpha 3.8$  1. Therefore, the deformation mechanism under the condition of strain rate of  $1.67 \times 10^{-4} s^{-1}$  is mispositioned viscose slip (solute drag creep) [10-16].

By this, you get the  $\alpha$  factor value under other conditions, and then construct a  $T$ - $\alpha$  three-dimensional  $\epsilon'$  bar chart, as shown in (Figure 1). It can be seen from the three-dimensional heliogram that at temperatures of 773 K and 793 K, with strain rates of  $1.67 \times 10^{-4} s^{-1}$ ,  $\alpha < 1$ , The speed control mechanism is the fault-climbing mechanism, and the stress index is  $n=5-4 s^{-1}$ , under the condition.  $\alpha > 2$  data  $\alpha > 1$ , speed control mechanisms are misplaced viscosity slip or solute drag creep, stress index  $n > 3$ ; Under other conditions, 14 data. The results of the bar chart determined by the mechanical data are [16] very consistent with the prediction of the deformation mechanism diagram in the literature. The introduction of  $\alpha$  determination factors and three-dimensional histograms makes the determination of the fault creep mechanism

simple and straightforward, and it is expected to help to deepen the understanding of the nature of creep.

## Conclusion

A criterion factor for determining the bit error viscosity slip (solute drag creep) and the climbing mechanism is proposed. The misalignment of the sliding mechanism is the dominant mechanism when  $\alpha$  is 1, and when  $\alpha < 1$ , the misalignment mechanism is the dominant mechanism. An example of calculation is given, and a three-dimensional histogram is constructed in the Al-1.44Mg-1.09Y alloy stretched at high temperature. The results show that under the conditions of temperature of 773 K and 793 K and strain rate of  $1.67 \times 10^{-4}$  The mechanism is misplaced climbing speed control  $s^{-1}$ , under the condition, the speed control mechanism is mispositioned viscosity slip or solute drag creep; Under other conditions, the misalignment [17-20].

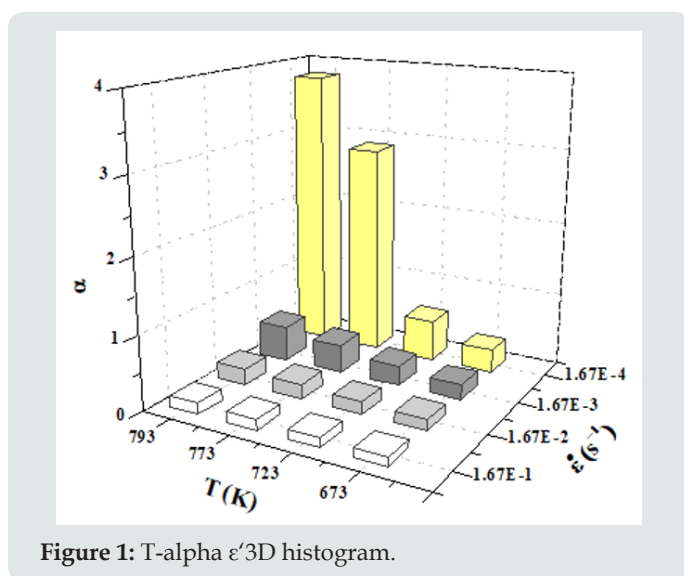


Figure 1: T- $\alpha$   $\dot{\epsilon}$  3D histogram.

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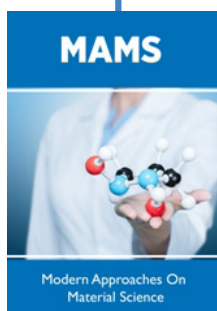
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